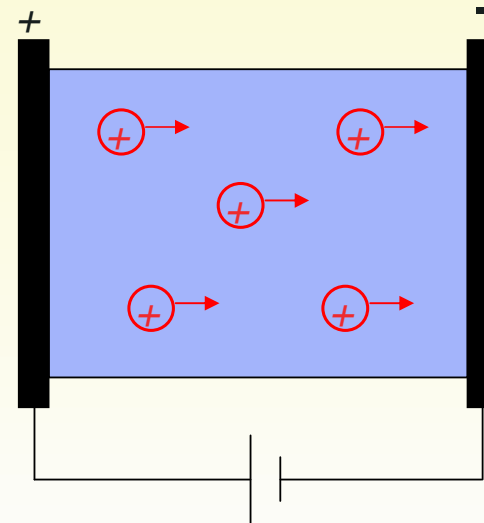
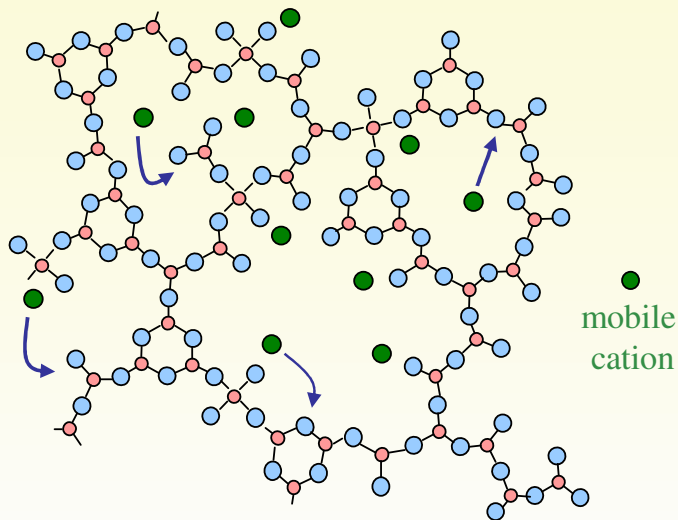


Ionically Conducting Materials: Bulk Ion Dynamics and Electrode Polarisation



Bernhard Roling

Department of Chemistry, University of Marburg

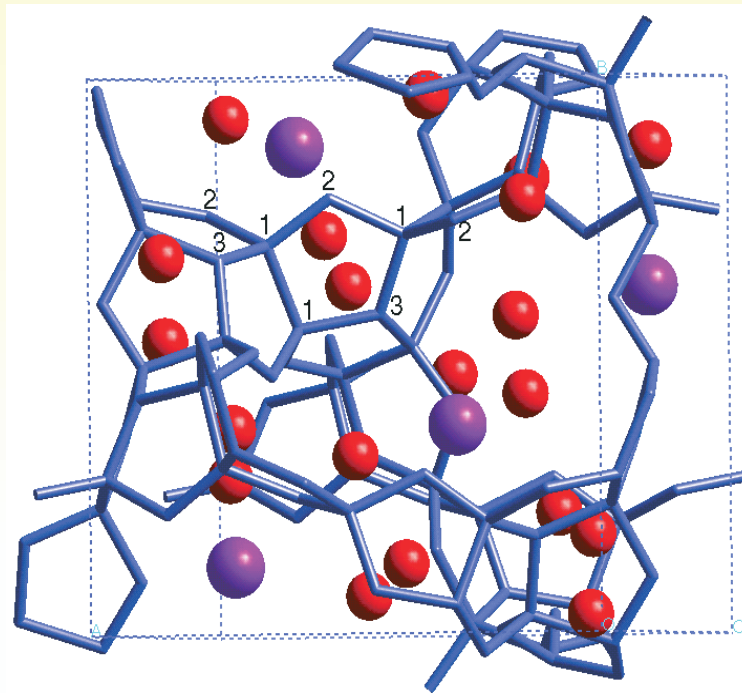
Outlook

1. Introduction about Ionically Conducting Materials
(Crystals, Glasses, Polymer Electrolytes, Ionic Liquids)
2. Simplest Equivalent Circuit for Ion Conductor between Blocking Electrodes:
Admittance, Impedance, Capacitance and Modulus Spectra
3. Bulk Ion Dynamics and Linear Response Theory
4. Double Layer Capacitance of Diluted and Concentrated Electrolytes
5. Conclusions

1. Introduction about Ionically Conducting Materials

Crystalline ion conductors

RbAg₄I₅: Ag⁺ ion conductor with $\sigma = 0,3$ S/cm at ambient temperatures



— Ag⁺ ion diffusion pathways

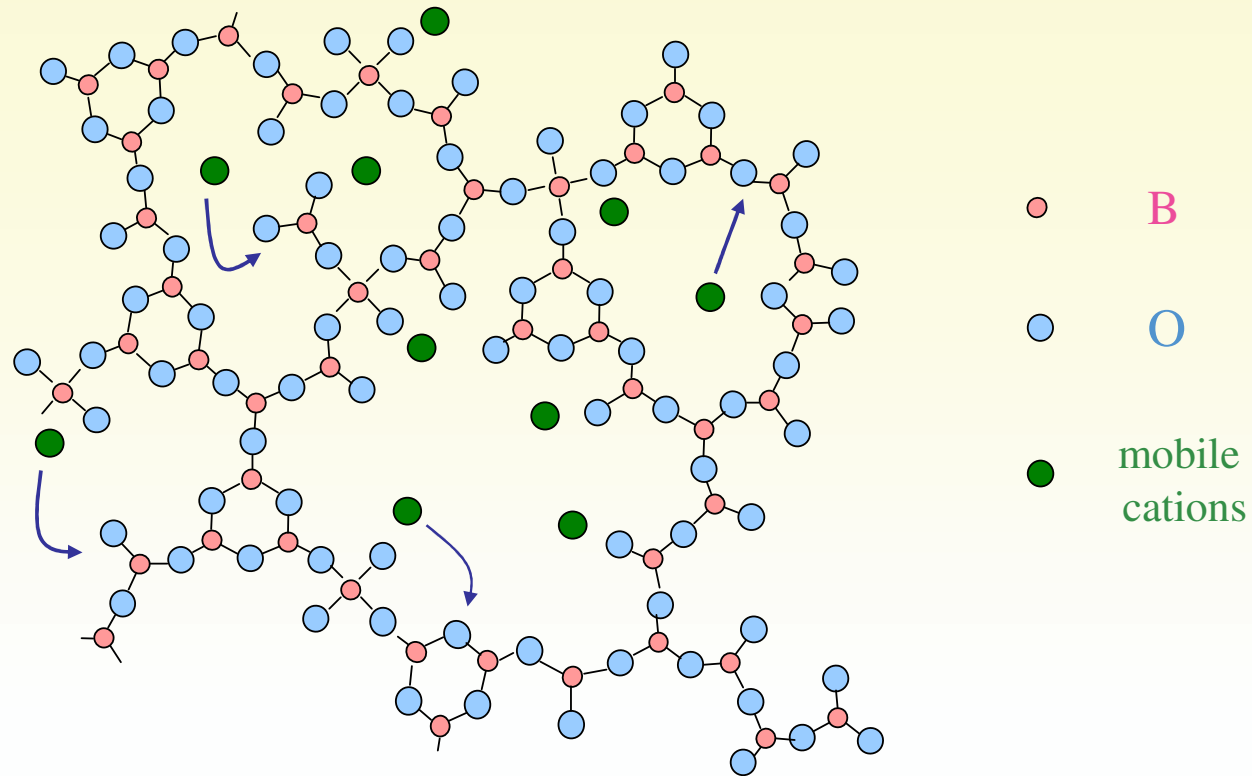
● Rb

● I

From: Stefan Adams,
Habilitation thesis,
Göttingen, 2000

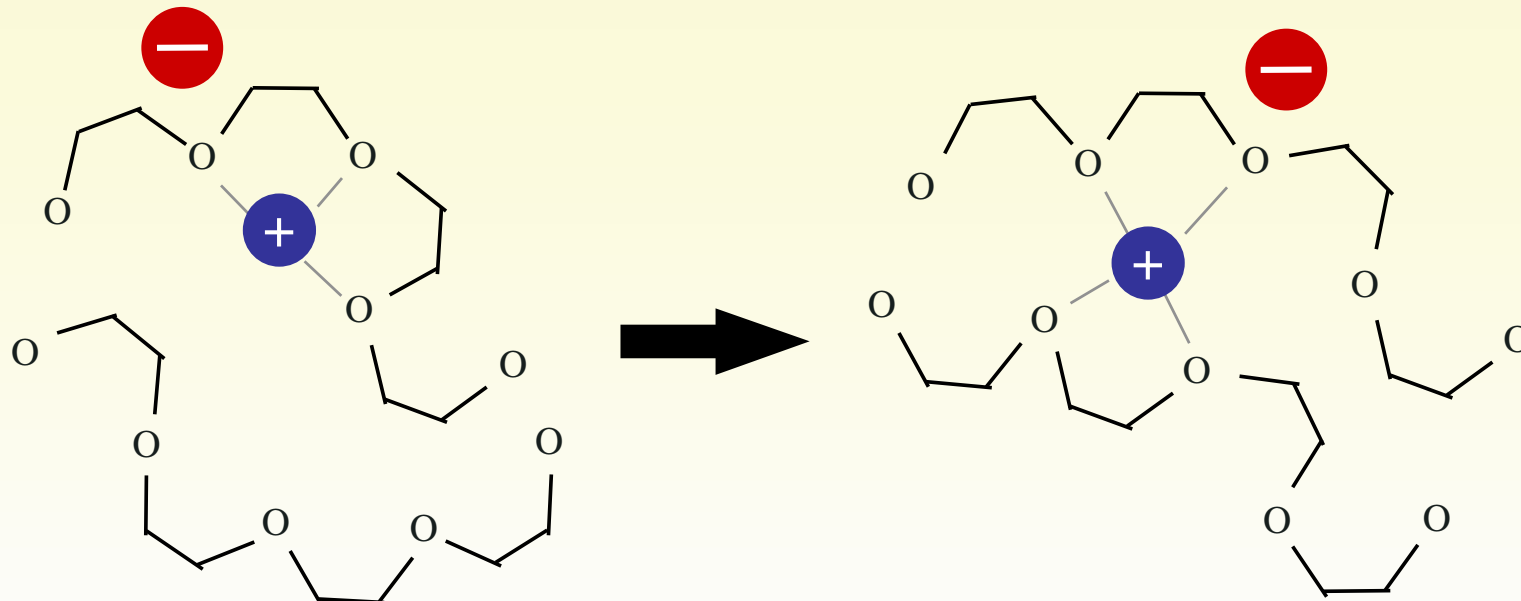
- Only 1/3 of all Ag⁺ sites are occupied by Ag⁺ ions.
- Weak chemical bonds between Ag⁺ ions und Rb/I matrix.

Ionically conducting glasses



- Ions move in a rigid amorphous matrix
- Disorder causes broad distribution of site energies and hopping barriers.

Amorphous salt-in-polymer electrolytes

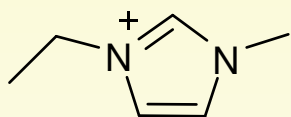


Polyethylene oxide +
salt, e.g. LiClO₄

Ion transport is assisted by
polymer segmental motion

Ionic Liquids

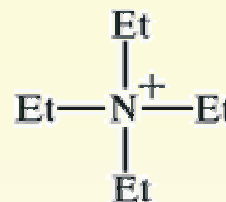
Typical Cations



Imidazolium cation

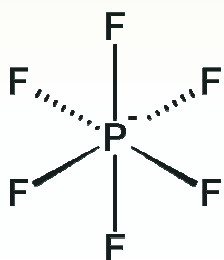


Pyrrolidinium cation



Tetraalkylammonium cation

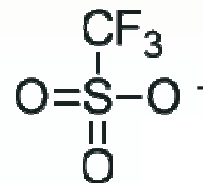
Typical Anions



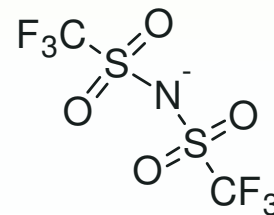
Hexafluorophosphate anion



Tetrafluoroborate anion



Triflate anion



Bis(trifluoromethanesulfonyl)imide anion (TFSI)

Important physico-chemical properties

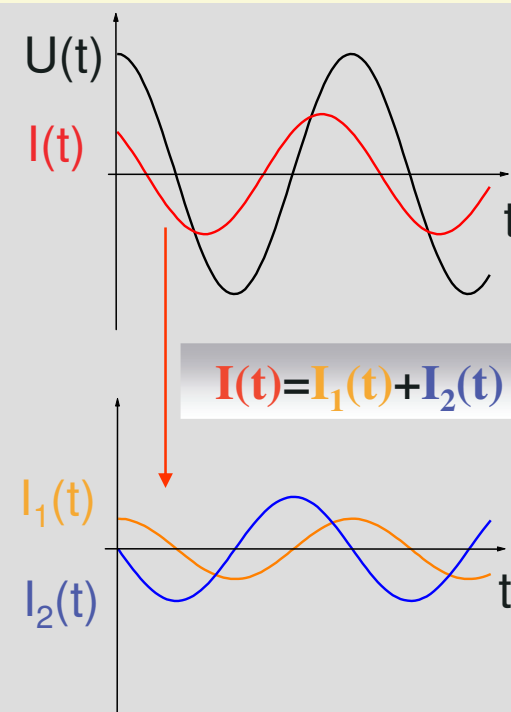
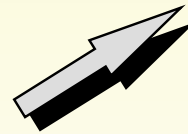
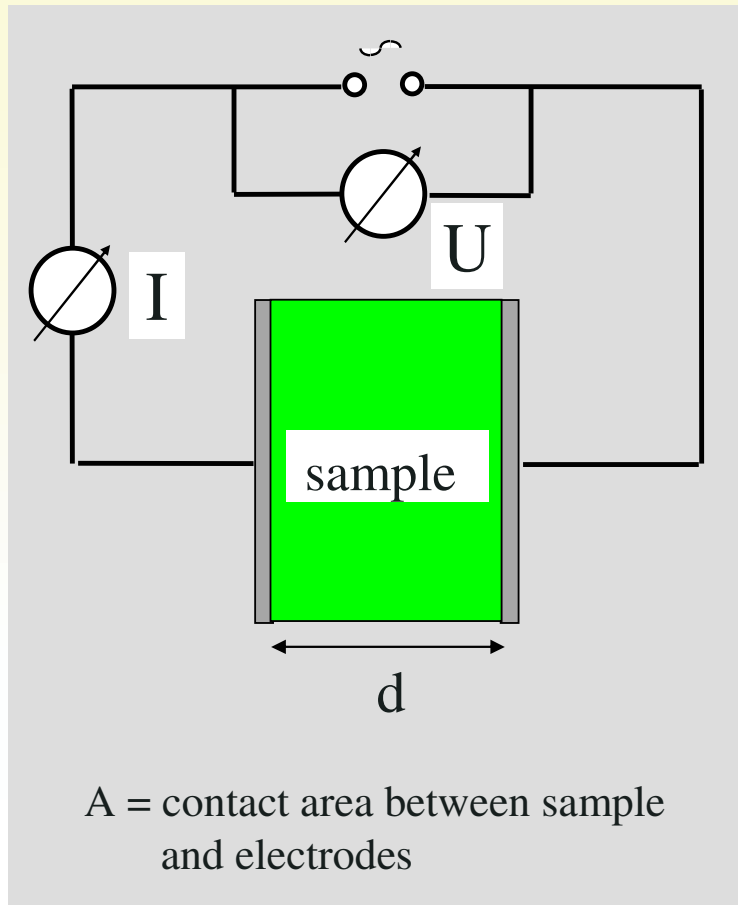
- low vapour pressure
- non-flammable
- high thermal stability
- high ionic conductivity ($10^{-3} - 10^{-2}$ S/cm at 298 K)
- broad electrochemical window (up to 5-6 V)



Application as electrolytes in batteries, supercapacitors,
fuel cells and dye-sensitized solar cells

- higher performance
- higher safety

2. Simplest Equivalent Circuit for Ion Conductor between Blocking Electrodes: Admittance, Impedance, Capacitance and Modulus Spectra



Admittance $\hat{Y}(\omega) = Y'(\omega) + iY''(\omega)$

$Y'(\omega) = I_{0,1} / U_0$ (Energy dissipation)

$Y''(\omega) = I_{0,2} / U_0$ (Energy storage)

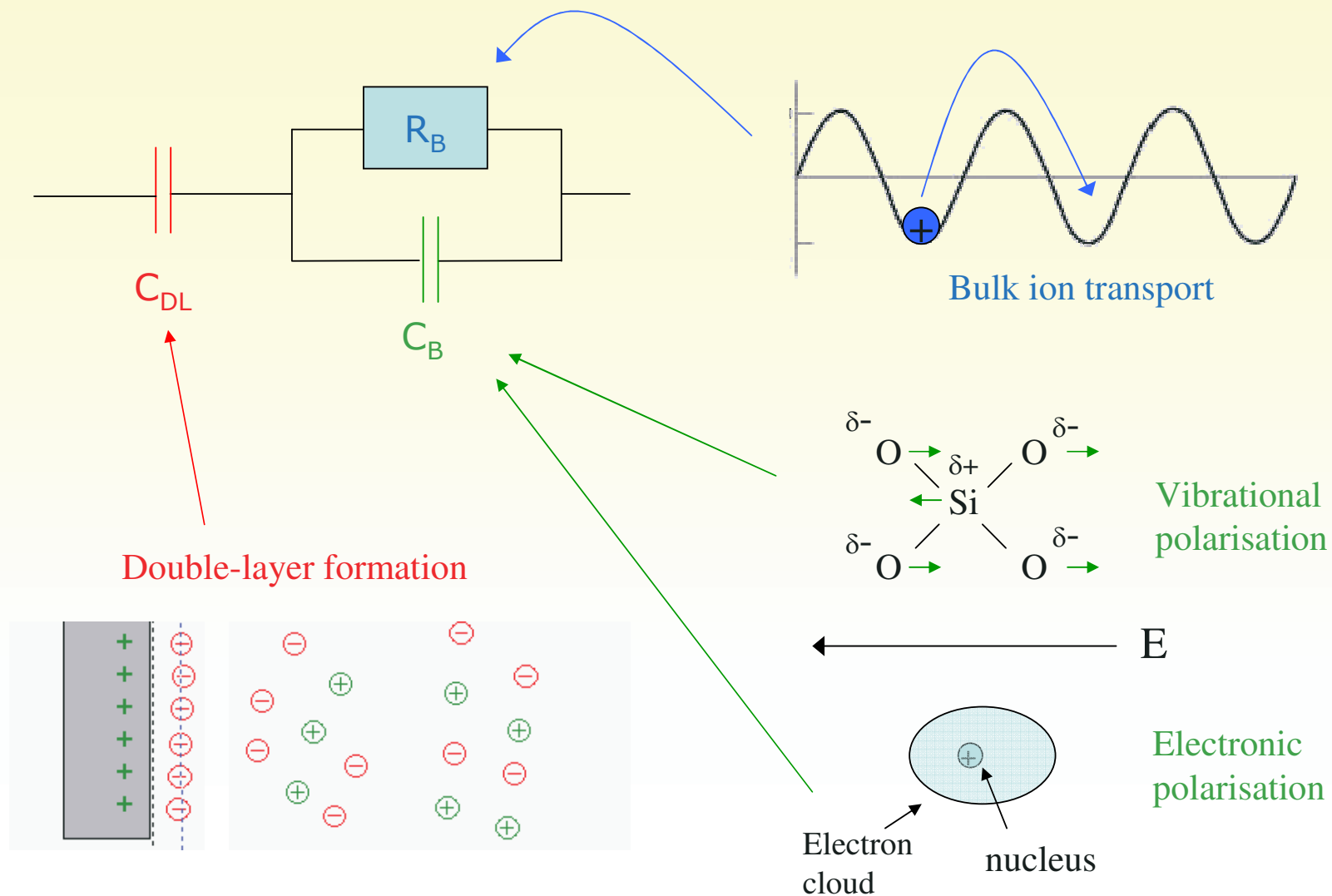
Other quantities

Impedance $\hat{Z}(\omega) = \frac{1}{\hat{Y}(\omega)} = Z'(\omega) + i Z''(\omega)$

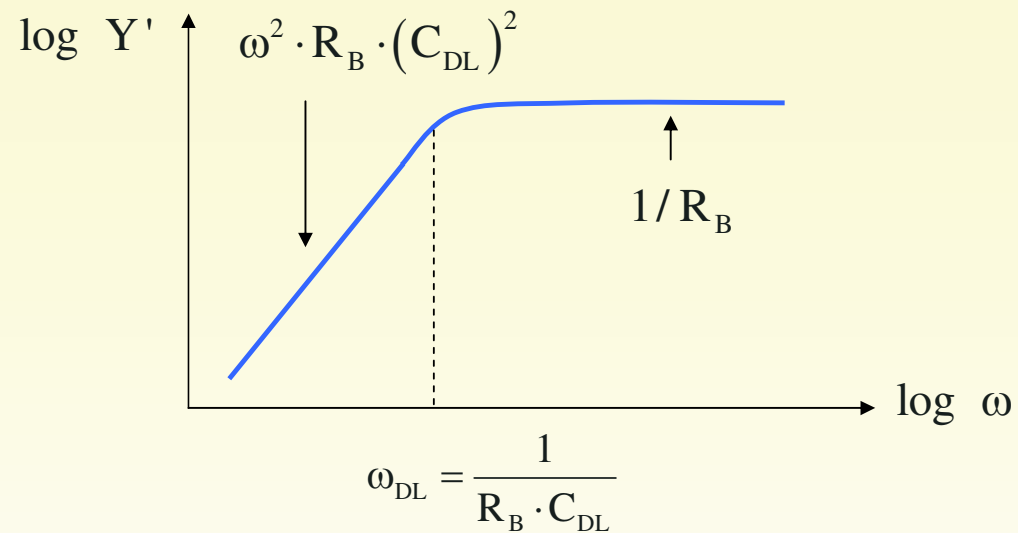
Capacitance $\hat{C}(\omega) = \frac{\hat{Y}(\omega)}{i\omega} = C'(\omega) - i C''(\omega)$

(Non-specific)
Modulus $\hat{M}(\omega) = \frac{1}{\hat{C}(\omega)} = M'(\omega) + i M''(\omega)$

Simplest circuit representing an ionic conductor between blocking electrodes



Real part of admittance

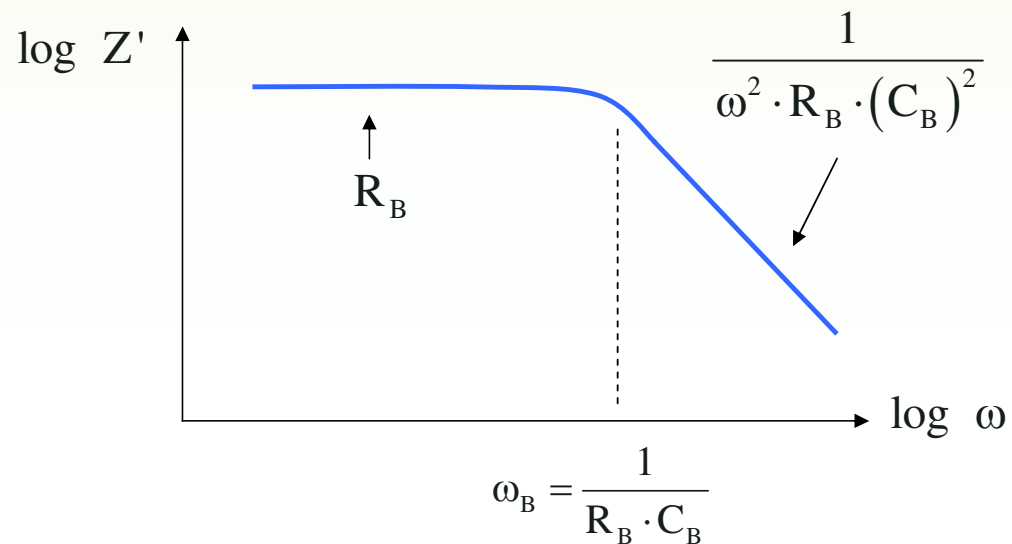


- Bulk ionic dc conductivity

$$\sigma_{dc} = \frac{1}{R_B} \cdot \frac{d}{A}$$

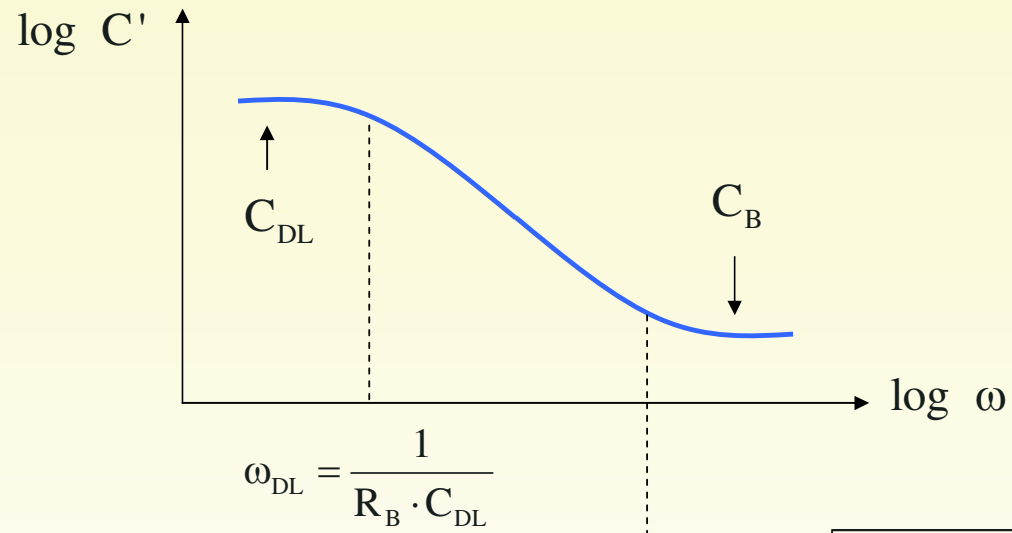
- No information about C_B

Real part of impedance

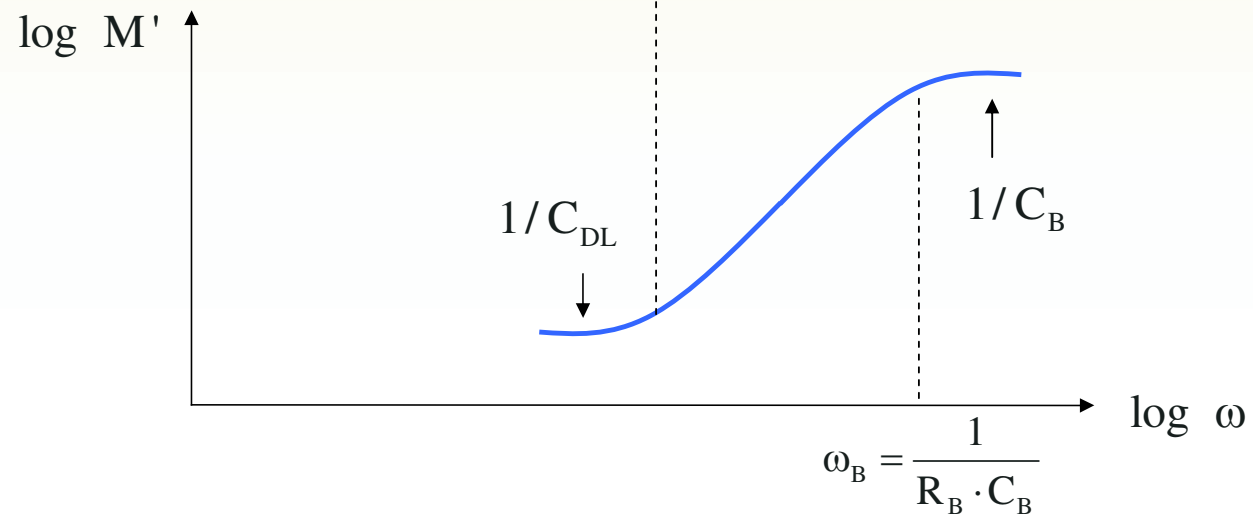


No information about C_{DL}

Real part of capacitance



Real part of modulus



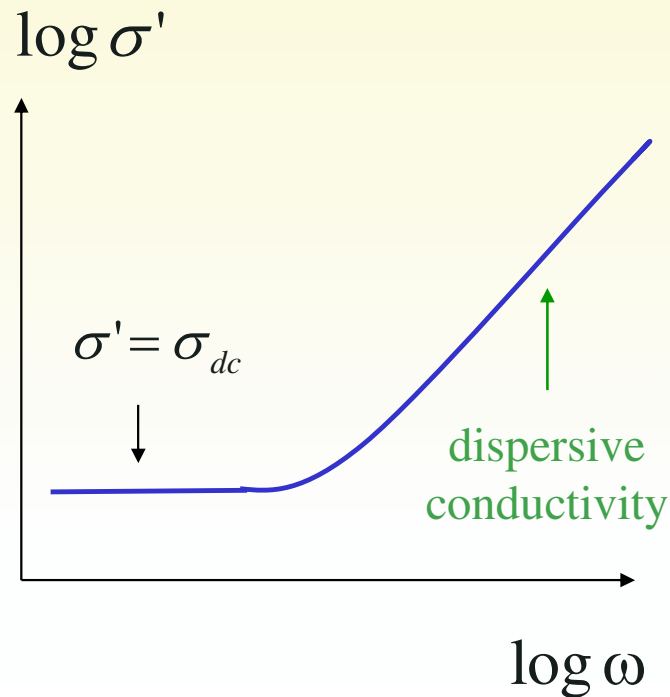
$$\omega_{avg} = \sqrt{\frac{1}{(R_B \cdot C_B) \cdot (R_B \cdot C_{DL})}} = \sqrt{\omega_B \cdot \omega_{DL}}$$

3. Bulk Ion Dynamics and Linear Response Theory

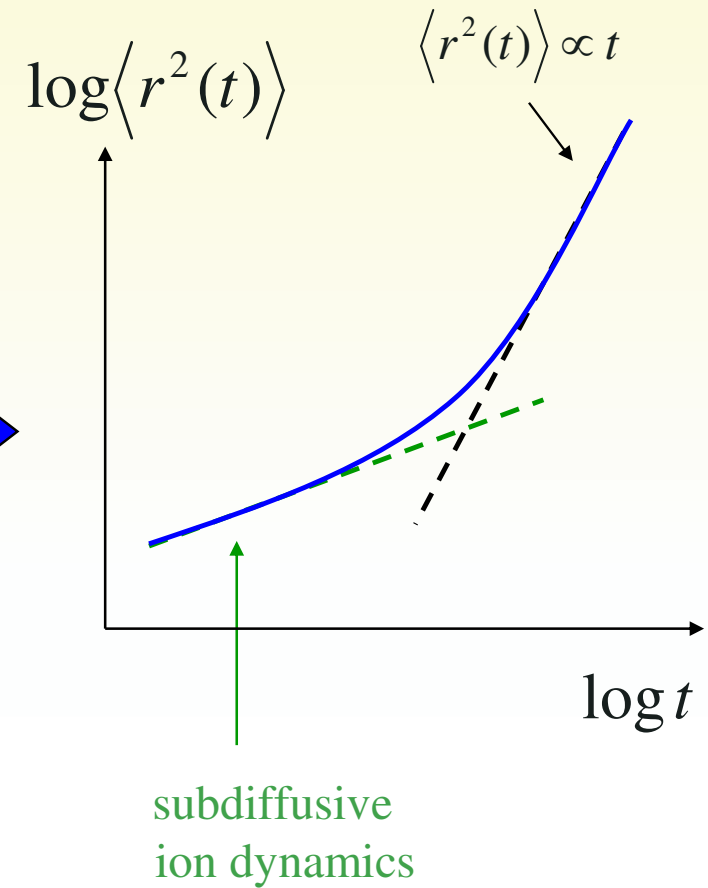
$$\sigma'(\omega) = Y'(\omega) \cdot \frac{d}{A}$$

$\langle r^2(t) \rangle$ = mean square displacement of ions

diffusive ion dynamics

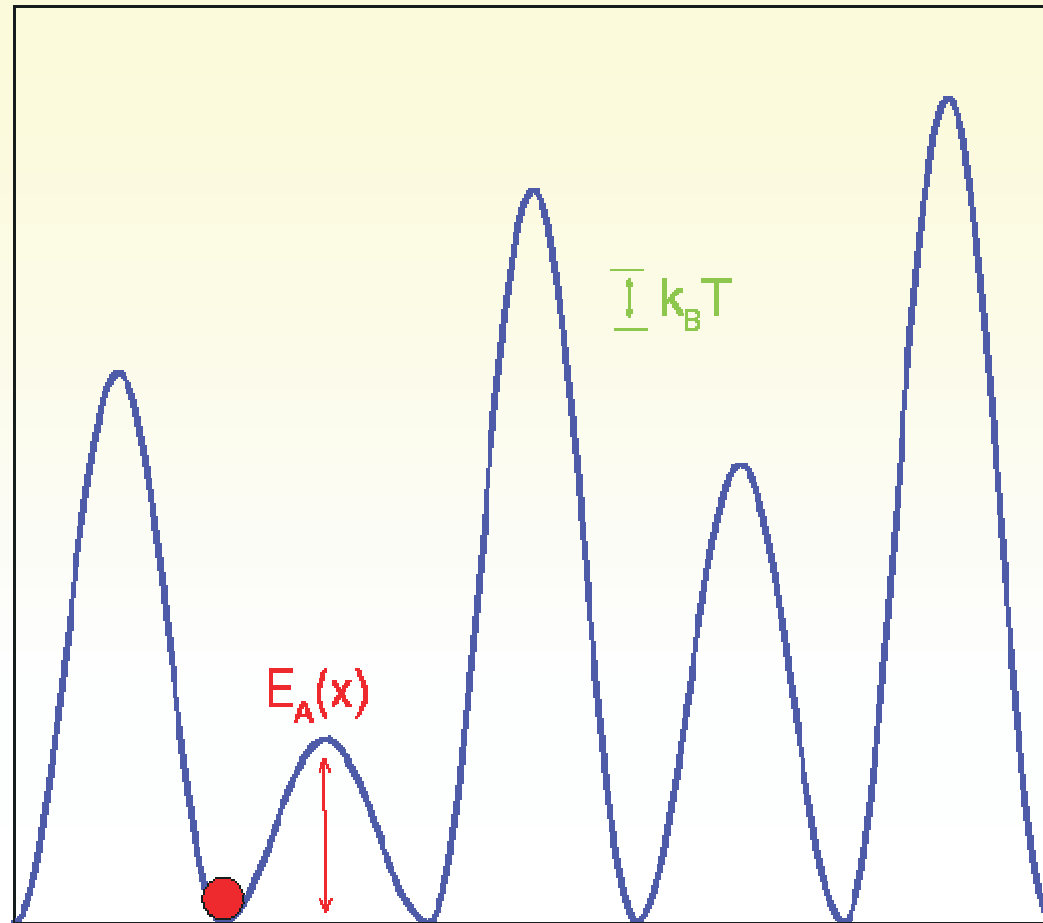


Linear response theory



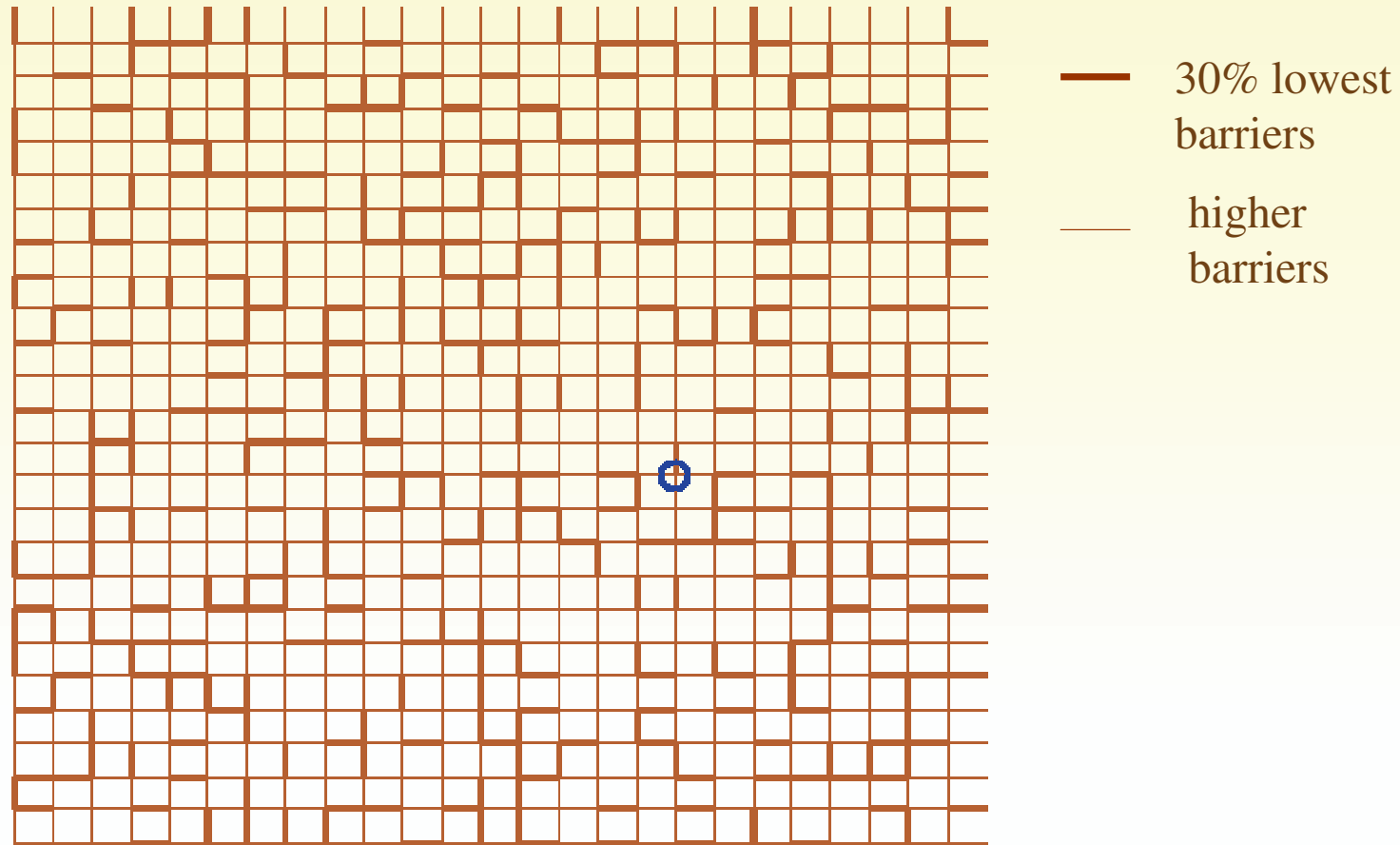
Random barrier model

Potential
energy
 V



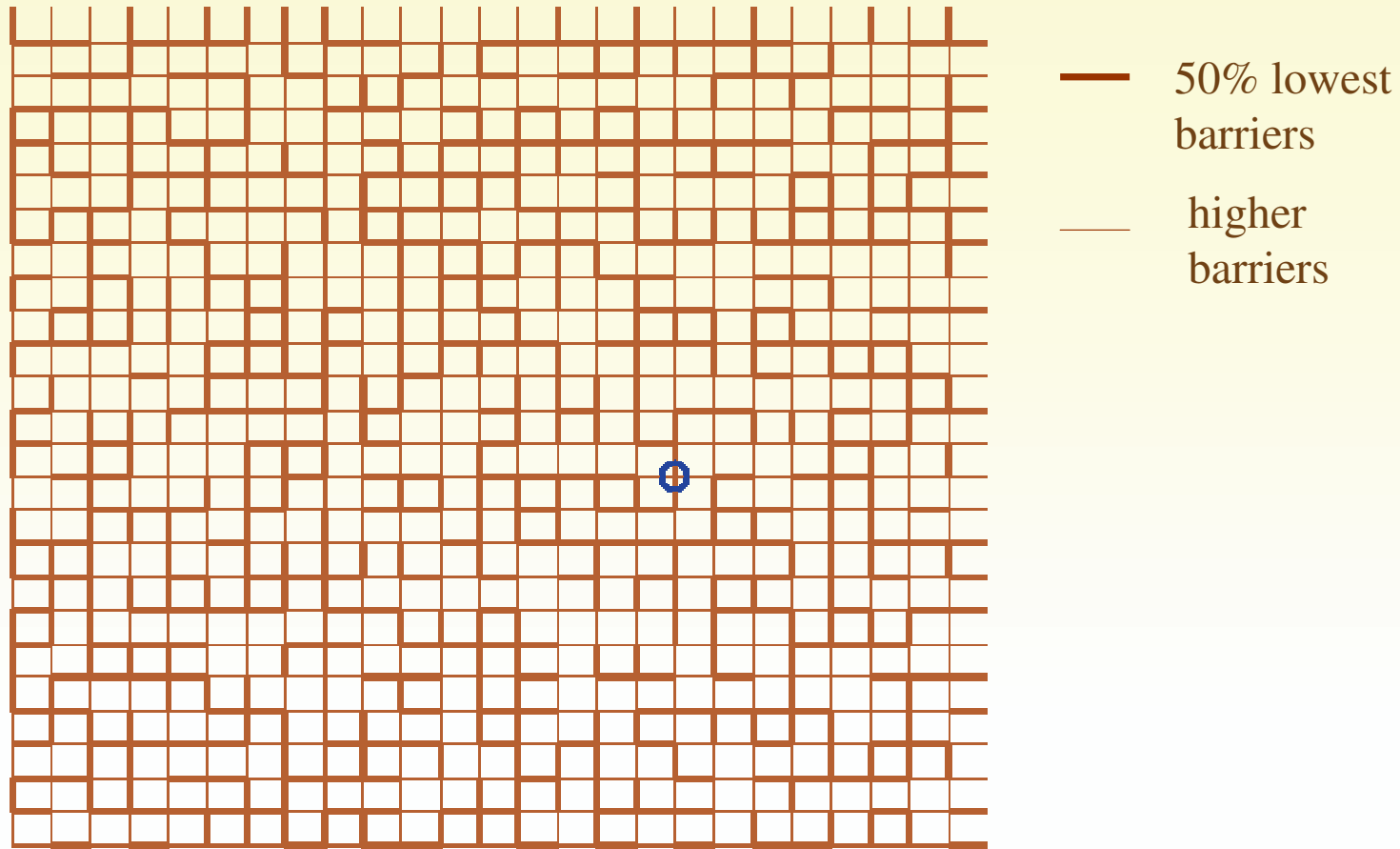
space coordinate x

Ion transport pathways in 2D random barrier model



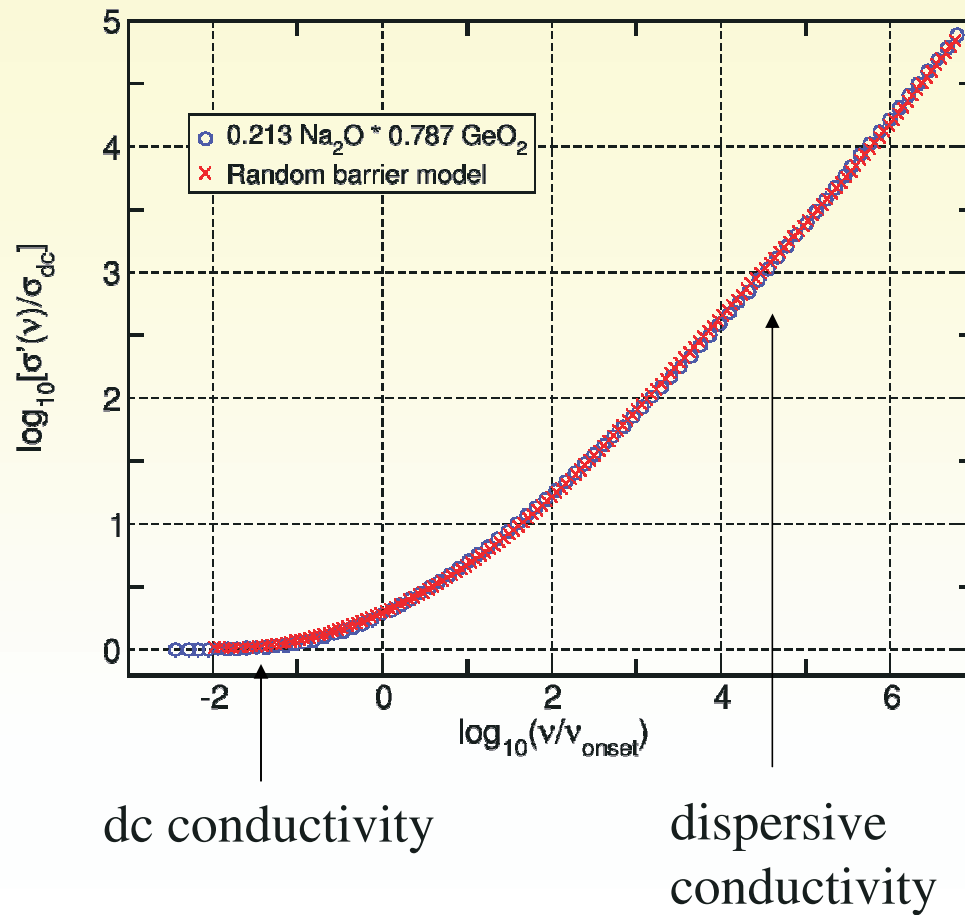
If the ions carry out hops exclusively over the 30% lowest barriers, the movements remain localised.

Ion transport pathways in 2D random barrier model



- Subdiffusive movements on short time scales
- Diffusive movements on long time scales.

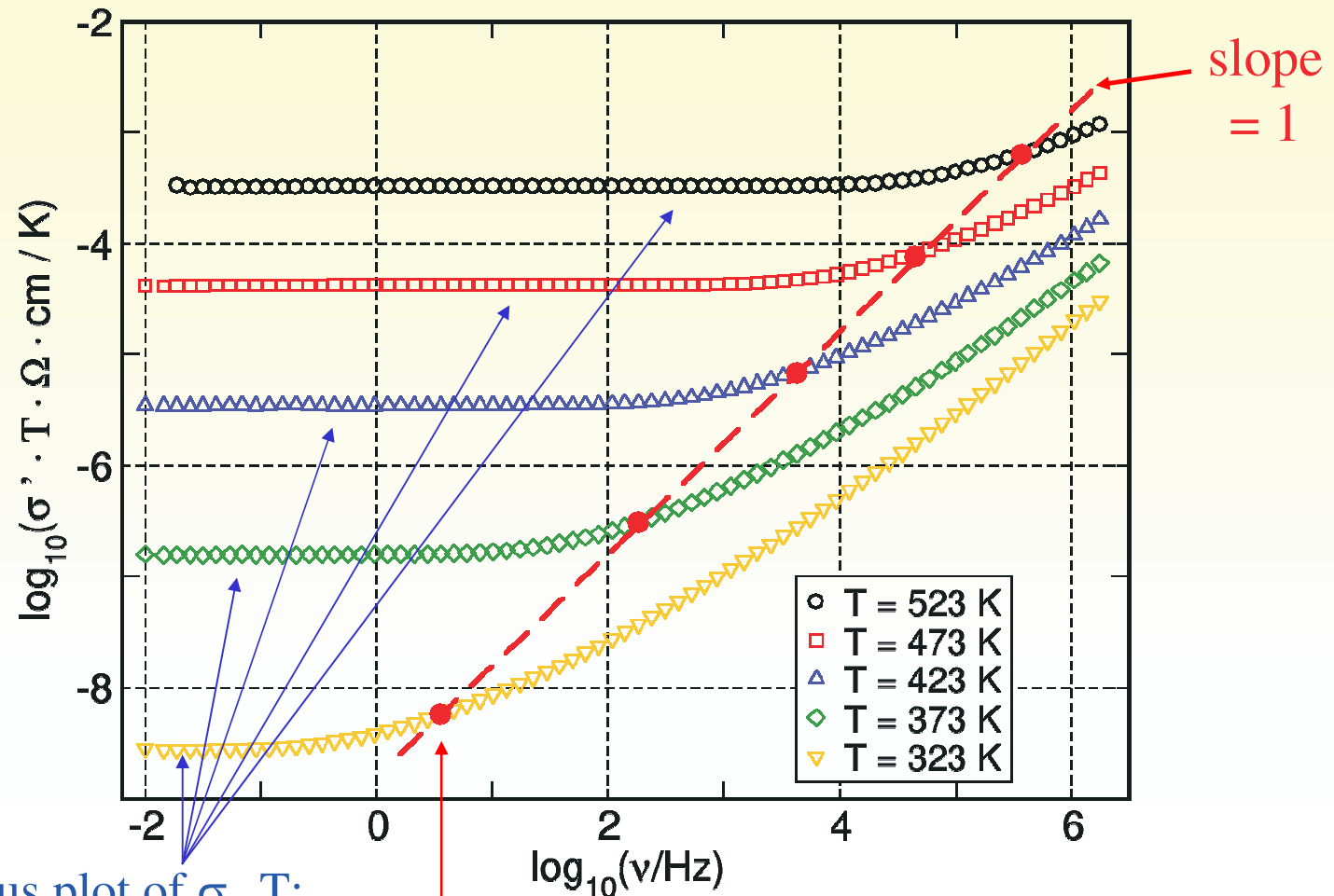
Comparison: Ion Conducting Glasses vs. Random Barrier Model



B. Roling,
*Phys. Chem.
Chem. Phys.*
3 (2001) 5093.

See also:
J. C. Dyre, T. Schroeder,
Rev. Mod. Phys.
72 (2000) 873.

Conductivity spectra of a 0.213 Na₂O * 0.787 GeO₂ glass

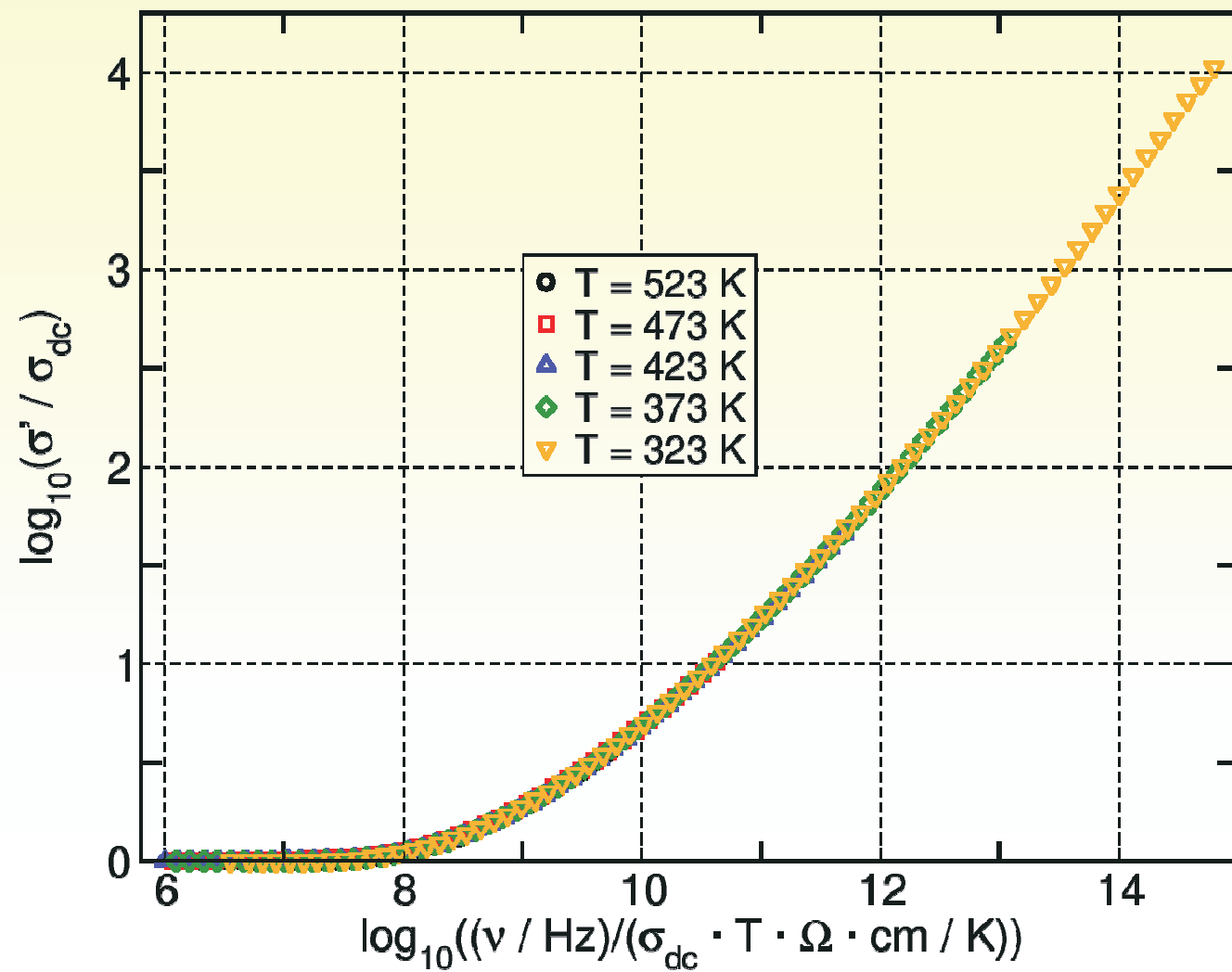


Arrhenius plot of $\sigma_{\text{dc}} T$:

Activation energy

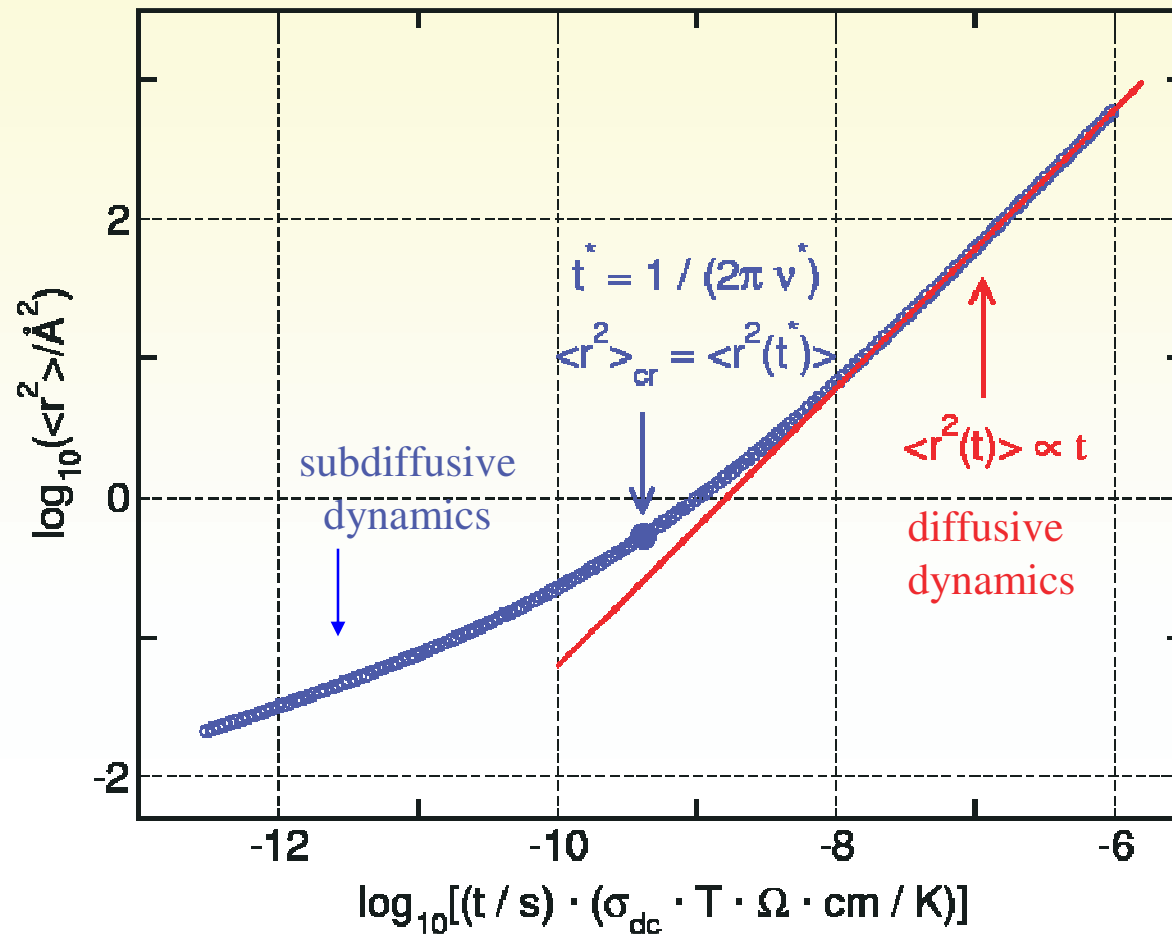
$$\sigma'(v^*) = 2 * \sigma_{\text{dc}}$$

Summerfield scaling



Master curve of the time-dependent mean square displacement of the mobile Na⁺ ions, $\langle r^2(t) \rangle$

0.213 Na₂O * 0.787 GeO₂ glass

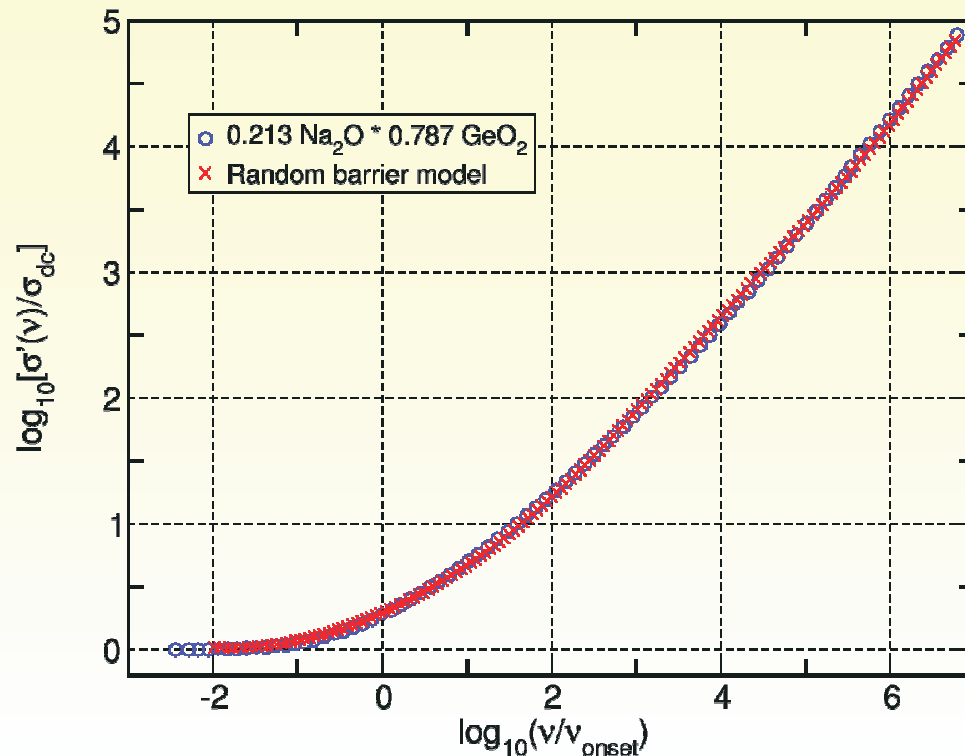


$$R \equiv \sqrt{\langle r^2 \rangle_{cr}}$$

“Spatial extent of
subdiffusive
dynamics”

For most glasses,
R is independent
of temperature.

Comparison: Ion Conducting Glasses vs. Random Barrier Model



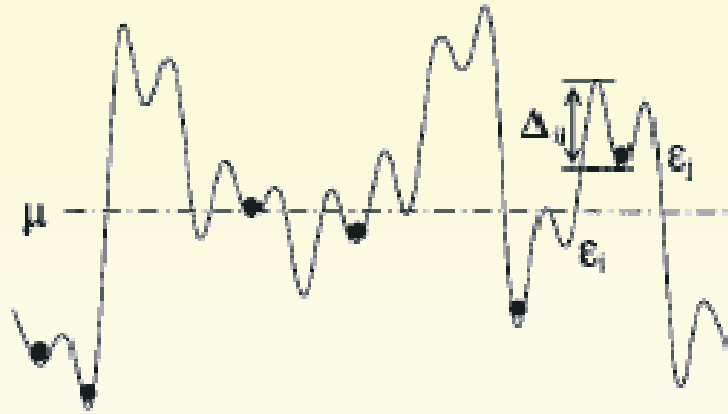
B. Roling,
Phys. Chem.
Chem. Phys.
3 (2001) 5093.

See also:
J. C. Dyre, T. Schroeder,
Rev. Mod. Phys.
72 (2000) 873.

However: Random Barrier Model: $\langle r^2 \rangle_{cr} \propto T^{-1.3}$ (Pathway structure depends on on temperature.)

Most ion conducting glasses: $\langle r^2 \rangle_{cr}$ independent of T

Random site-energy model with hard-core interactions between particles



S. D. Baranovskii,
H. Cordes,
J. Chem. Phys.
111 (1999) 7546

Empty site can only be occupied by a single ion. $\longrightarrow$ Fermi statistics

$$\longrightarrow \langle r^2 \rangle_{cr} \propto T^{-0.3}$$

The number of mobile ions that can hop to an empty site close to the Fermi level is proportional to T .

Random barrier model with Coulomb interactions between mobile ions

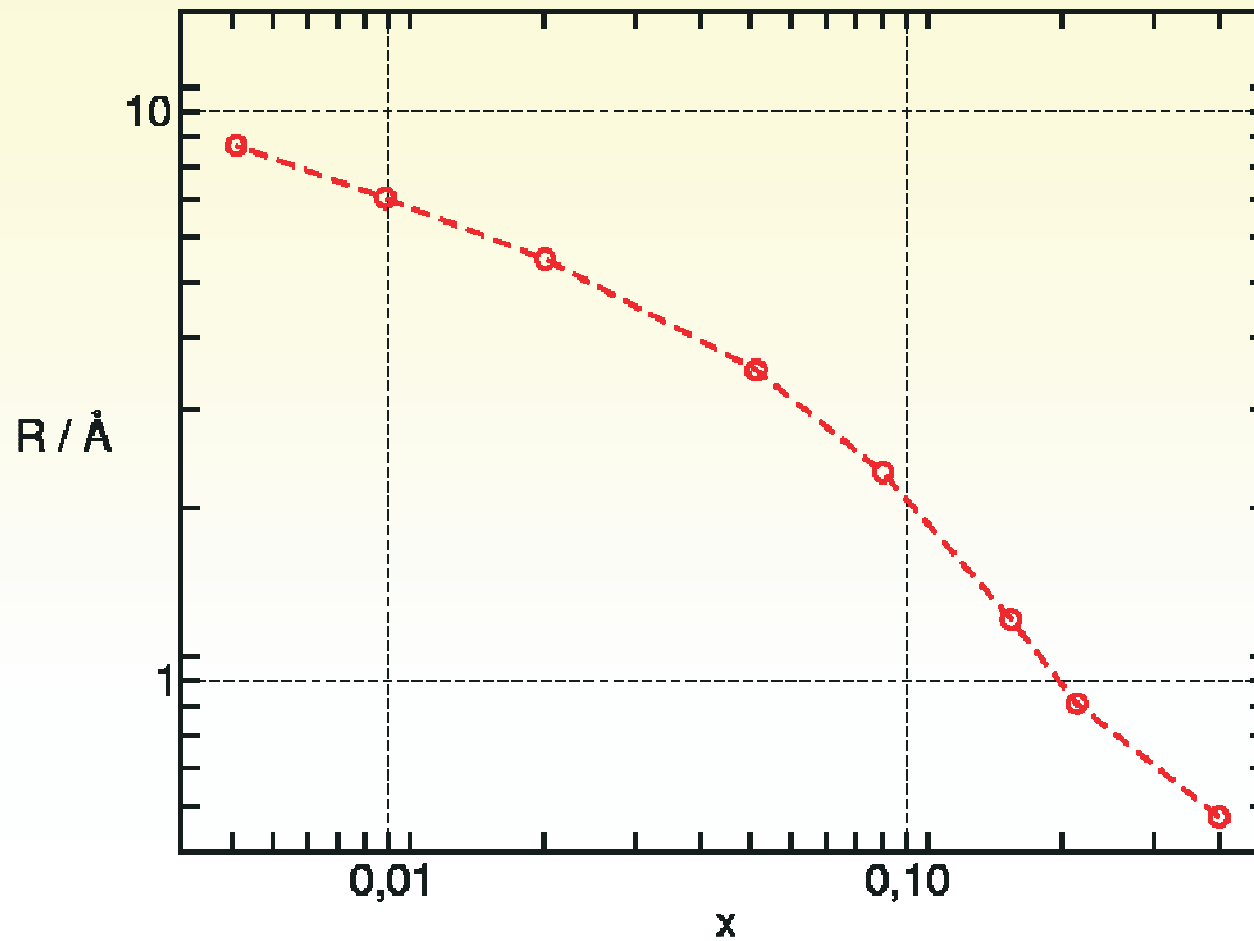
Monte Carlo simulations:

$$\left\langle r^2 \right\rangle_{cr} \propto T^\gamma$$

- without Coulomb interactions: $\gamma = -1.3$
- with increasing strength of Coulomb interaction:
 γ increases (i.e. becomes more positive)

B. Roling, *Phys. Chem. Chem. Phys.* **3** (2001) 5093

Spatial extent of subdiffusive ion dynamics $R \equiv \sqrt{\langle r^2 \rangle_{\text{cr}}}$
 for $x \text{ Na}_2\text{O} * (1 - x) \text{ GeO}_2$ glasses



B. Roling, C. Martiny, S. Brückner, *Phys. Rev. B* **63** (2001) 214203.

Assumption: Typical hopping distance of Na^+ ions: $d \approx 3 \text{ \AA}$
(Molecular dynamics simulations)

Glass with $x = 0.005$: $R > d$

→ At the crossover time t^* , the ions have moved, on the average, over *several hopping distances*.

Glass with $x = 0.40$: $R < d$

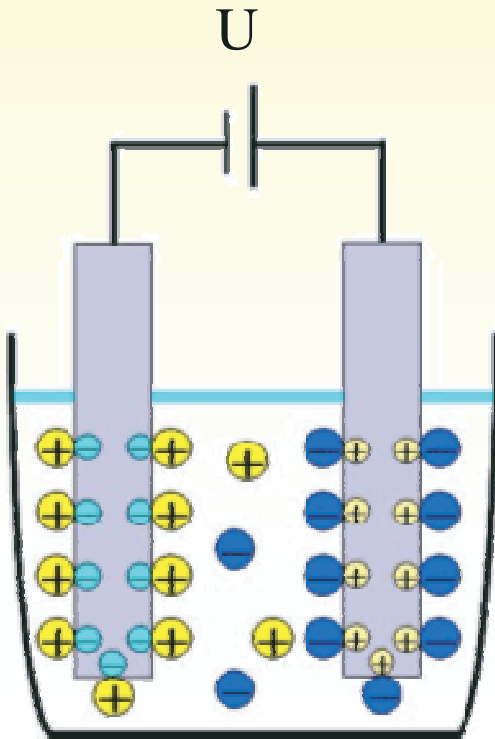
→ At the crossover time t^* , only a *small fraction of ions* have left their original sites.



Despite the quasi-universal shape of the conductivity spectra, the microscopic mechanisms of the ion transport depend strongly on glass composition.

4. Double Layer Capacitance of Diluted and Concentrated Electrolytes

Energy storage in electrochemical supercapacitors



Stored energy

$$E = \frac{1}{2} \cdot C \cdot U^2$$

C = double layer capacitance

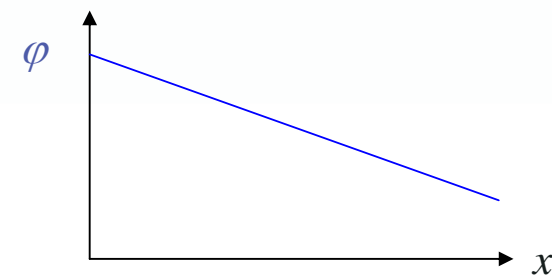
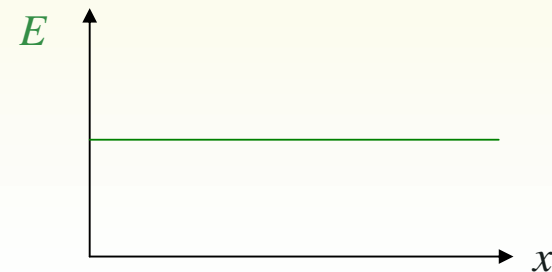
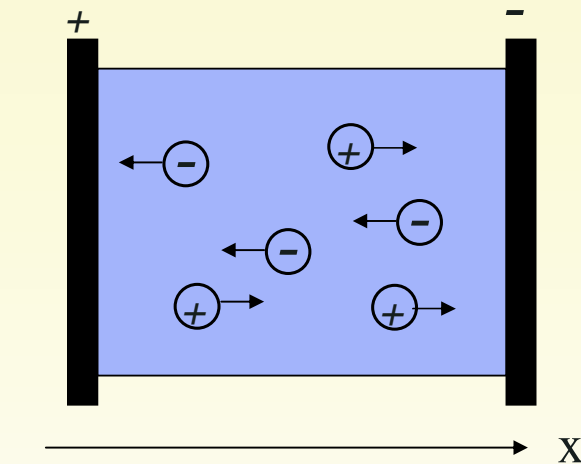
U = voltage

Power

$$P = \frac{U^2}{R}$$

R = electrolyte resistance

Mean-field approaches for diluted electrolyte solutions



Charge density

$$\rho(x) = \frac{Q(x)}{V} =$$

$$N_{v,+}(x) \cdot z_+ \cdot e + N_{v,-}(x) \cdot z_- \cdot e = 0$$



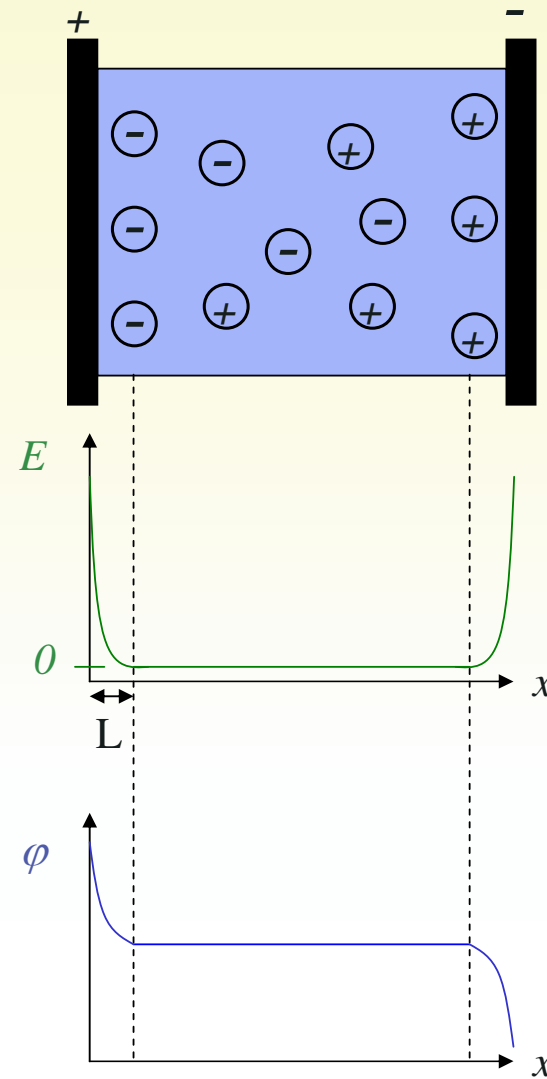
Poisson equation

$$\frac{dE(x)}{dx} = \frac{\rho(x)}{\epsilon_0 \cdot \epsilon_r} = 0$$

ϵ_0 = vacuum permittivity

ϵ_r = relative permittivity

$$\frac{d^2\phi(x)}{dx^2} = -\frac{\rho(x)}{\epsilon_0 \cdot \epsilon_r} = 0$$



close to the electrodes:

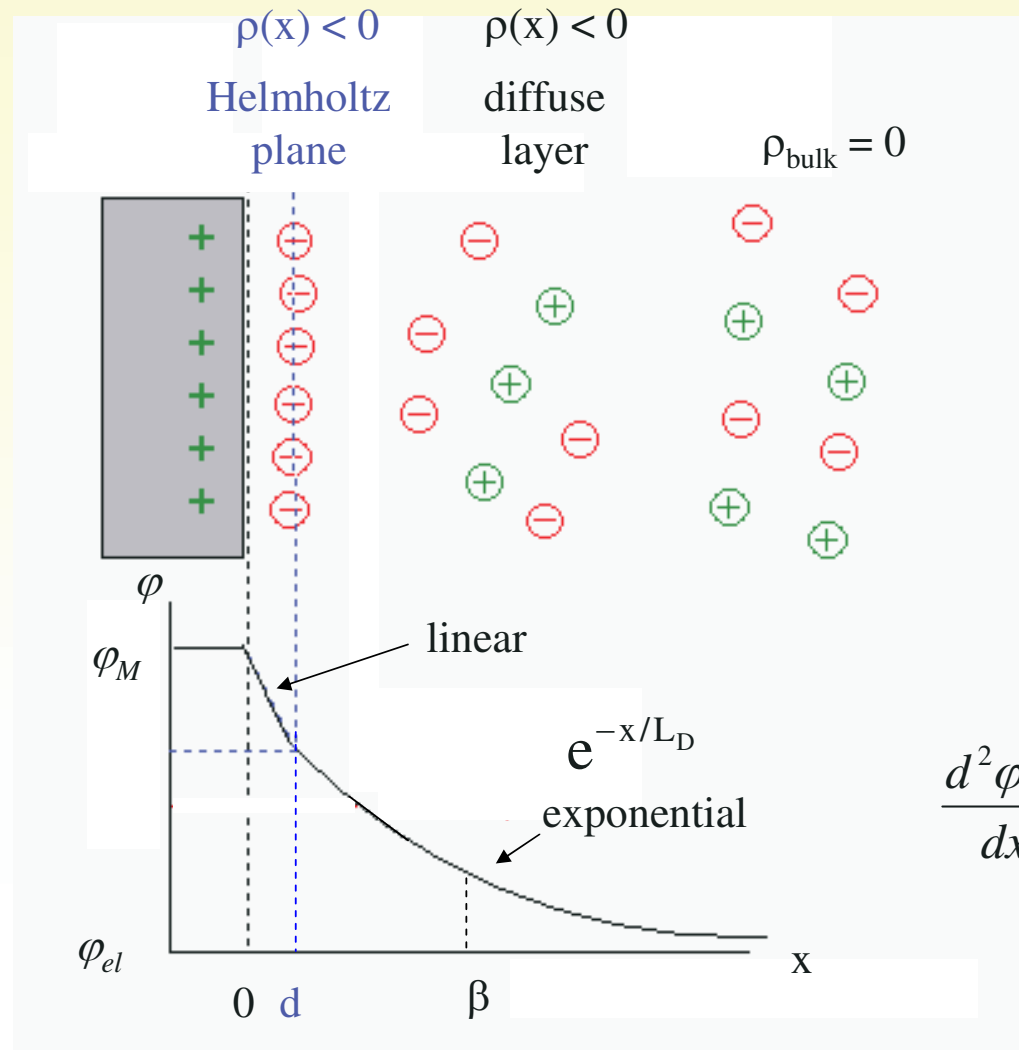
$$\rho(x) \neq 0$$

in the bulk: $\rho(x) = 0$

$$\frac{dE(x)}{dx} = \frac{\rho(x)}{\epsilon_0 \cdot \epsilon_r}$$

$$\frac{d^2\phi(x)}{dx^2} = -\frac{\rho(x)}{\epsilon_0 \cdot \epsilon_r}$$

Now in more detail....



$$\frac{d^2\varphi(x)}{dx^2} = -\frac{\rho(x)}{\varepsilon_0 \cdot \varepsilon_r}$$

Capacitance of the double layer

Integral capacitance per area $K_A = \frac{C}{A} = \frac{Q / A}{\Delta\varphi}$ with $\Delta\varphi = \varphi_M - \varphi_{el}$

Helmholtz layer $K_A^{\text{HL}} = \frac{\varepsilon_0 \cdot \varepsilon_r}{d}$

Diffuse layer $K_A^{\text{diffuse}} = \dots$ (see next slides)



$$\frac{1}{K_A} = \frac{1}{K_A^{\text{HL}}} + \frac{1}{K_A^{\text{diffuse}}}$$

Gouy-Chapman theory for diffuse layer

Poisson
$$\frac{d^2\varphi(x)}{dx^2} = -\frac{\rho(x)}{\varepsilon_0 \cdot \varepsilon_r}$$

Boltzmann
$$\rho(x) = N_{V,+,bulk} \cdot z_+ \cdot e \cdot \exp\left(-\frac{z_+ \cdot e \cdot [\varphi(x) - \varphi_{el}]}{kT}\right) + N_{V,-,bulk} \cdot z_- \cdot e \cdot \exp\left(-\frac{z_- \cdot e \cdot [\varphi(x) - \varphi_{el}]}{kT}\right)$$

mean-field approach

Solution for 1-1 electrolyte with $z_+ = 1$, $z_- = -1$, $N_V = N_{V,+,bulk} = N_{V,-,bulk}$

$$\frac{Q}{A} = \frac{\varepsilon_0 \cdot \varepsilon_r \cdot 2 \cdot kT}{e \cdot L_D} \cdot \sinh\left(\frac{e \cdot \Delta\varphi}{2 \cdot kT}\right)$$

with
$$L_D = \sqrt{\frac{\varepsilon_0 \cdot \varepsilon_r \cdot kT}{2 \cdot N_V \cdot e^2}}$$
 Debye length

1-1 electrolyte at **small** potential differences $\Delta\phi$

$$e \cdot \Delta\phi < kT \quad \longrightarrow \quad \sinh\left(\frac{e \cdot \Delta\phi}{2 \cdot kT}\right) \approx \frac{e \cdot \Delta\phi}{2 \cdot kT}$$

$$\longrightarrow \quad \frac{Q}{A} = \frac{\epsilon_0 \cdot \epsilon_r}{L_D} \cdot \Delta\phi$$

$$K_A^{\text{diffuse}} = \frac{\epsilon_0 \cdot \epsilon_r}{L_D} \equiv C_D$$

Debye capacitance

for
 $\Delta\phi < 0.025 \text{ eV}$
at 25°C

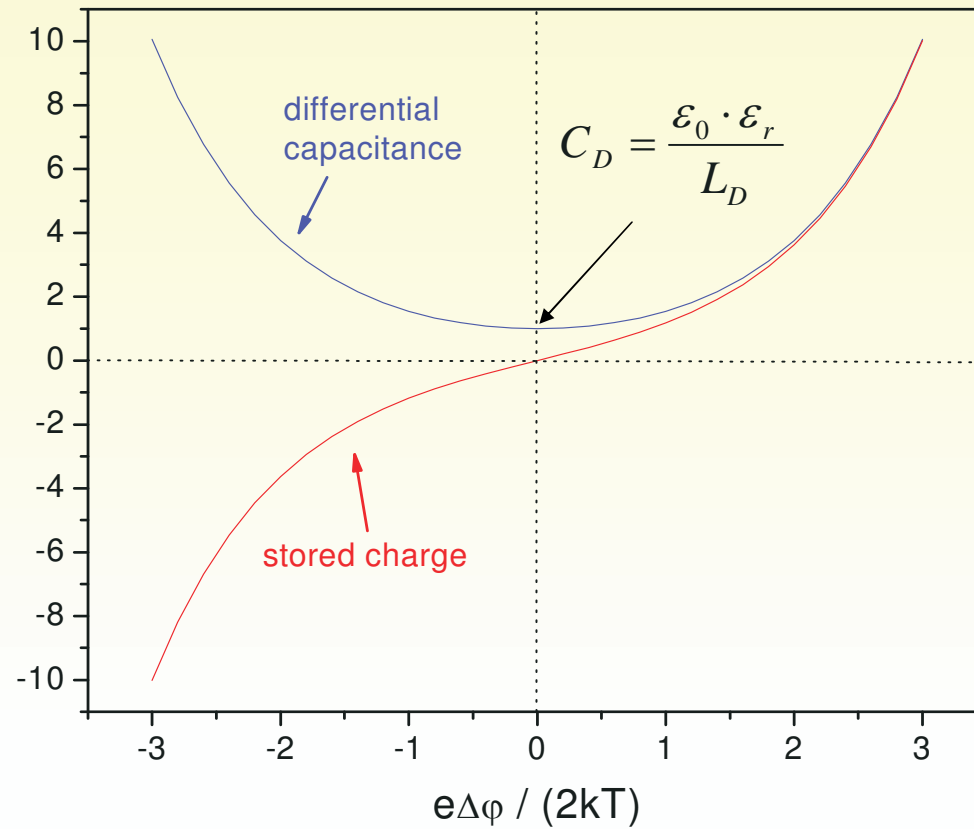
1-1 electrolyte at **large** potential differences $\Delta\phi$

$$\frac{Q}{A} = \frac{\epsilon_0 \cdot \epsilon_r \cdot 2 \cdot kT}{e \cdot L_D} \cdot \sinh\left(\frac{e \cdot \Delta\phi}{2 \cdot kT}\right)$$

→ Differential capacitance

$$C_A^{\text{diffuse}} \equiv \frac{d(Q/A)}{d(\Delta\phi)} = \frac{\epsilon_0 \cdot \epsilon_r}{L_D} \cdot \cosh\left(\frac{e \cdot \Delta\phi}{2 \cdot kT}\right) = C_D \cdot \cosh\left(\frac{e \cdot \Delta\phi}{2 \cdot kT}\right)$$

Stored charge and differential capacitance



At highly positive and negative electrode potentials:
Strong accumulation of counter ions at the electrode.

Double layer formation in ionic liquids

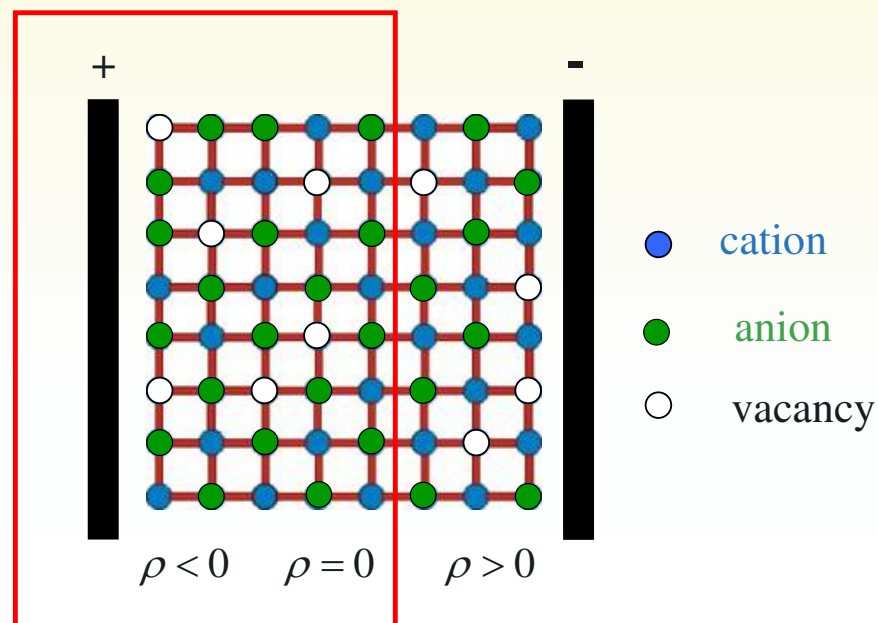
Gouy-Chapman theory NOT applicable:

In concentrated systems, the finite volume of the ions and the individual interactions between the ions play an important role.

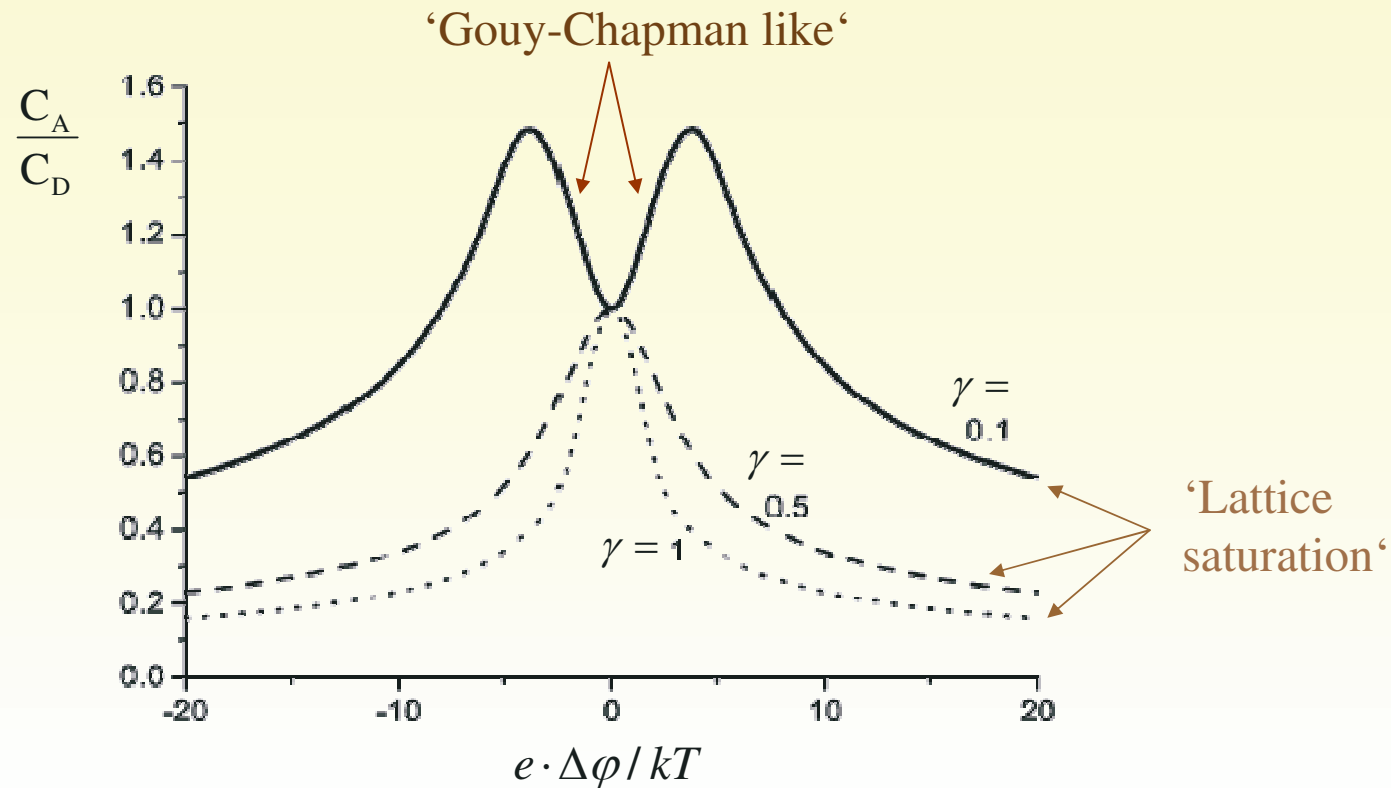
A. A. Kornyshev, *J. Phys. Chem. B* **111** (2007) 5545-5557.

Mean-field lattice gas model

- Hard-core interactions between ions are implicitly taken into account.
- Individual Coulomb interactions are neglected.
- Model parameters:
 - (i) $e \cdot \Delta\phi / kT$
 - (ii) ratio of occupied sites
 $\gamma = 0 \dots 1$



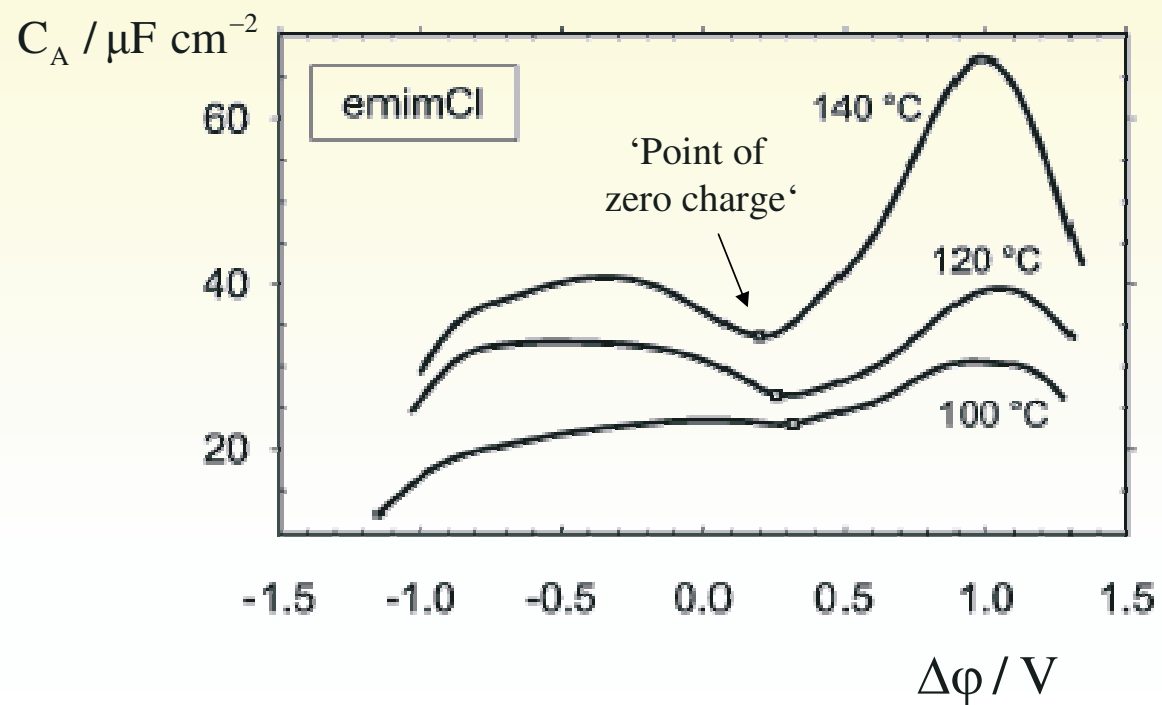
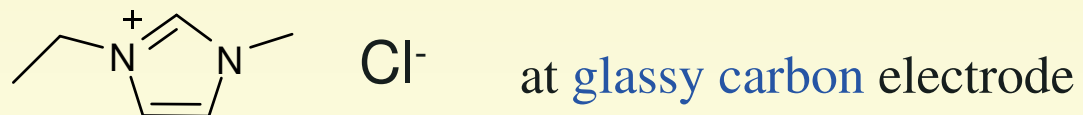
Kornyshev's results for differential capacitance of double layer



At highly positive and negative electrode potentials:
Since ion concentration at the electrodes cannot become much higher than in the bulk,
the thickness of the double layer has to increase with increasing potential.

→ 'Lattice saturation'

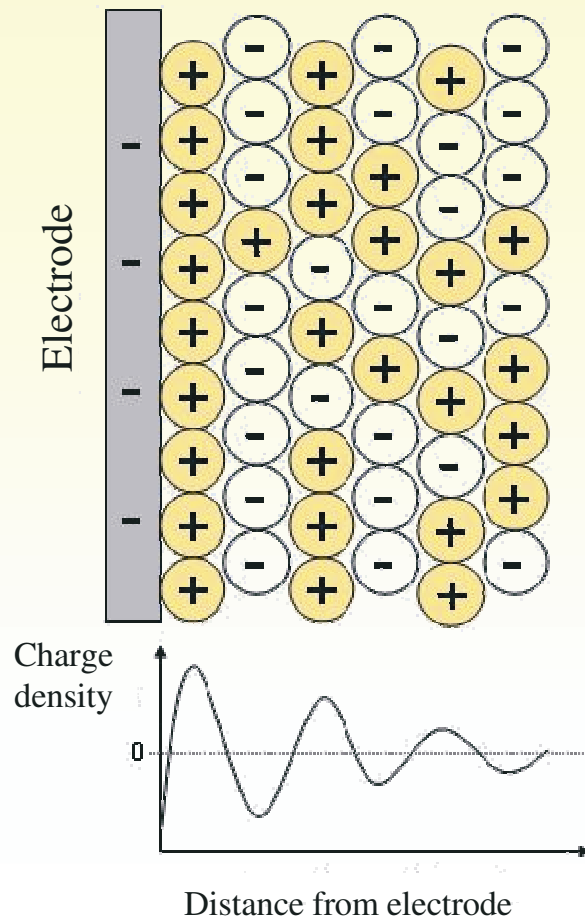
Experimental results for double layer capacitance



Lockett et al.,
J. Phys. Chem. C
112 (2008) 7486.

Similar to Kornyshev's results for $\gamma = 0.1$!

Structure of Double Layer from Molecular Dynamics Simulations



Charge density oscillations and 'Overscreening effect':

“Too much” positive charge in the first layer at the negative electrode.

Seems to be a universal interfacial phenomenon in dense ionic systems.

However: Influence on differential capacitance not yet well understood.

5. Conclusions

Bulk ion dynamics:

- Vast amount of experimental data
- Many well established models
- However, no model can describe all experimental features.
- Still, many basic experimental results are not fully understood.

Review article on open questions in this field: J. C. Dyre, P. Maass, B. Roling, D. L. Sidebottom, *Rep. Prog. Phys.* **72** (2009) 046501.

Electrode polarisation in concentrated electrolytes (Ionic liquids and solid electrolytes)

- Reliable experimental data are scarce.
(Impurities may have strong impact on results!)
- Currently, new model approaches are being developed and sophisticated molecular dynamics simulations are being carried out.
- Apart from purely electrostatic double layer charge storage, pseudocapacitive processes play an important role.

Talk about electrode polarisation in ILs: Ionic Solids Session, tomorrow, 12.00-12.30